



## RESEARCH ARTICLE

### ANALYSIS AND DESIGN OF COOLING CHAMBER FOR GAS ATOMIZATION PLANT

\*Pankaj Sharma

Department of Mechanical Engineering, Dr B R A College of Agril Engg & Tech, Etawah – 206001 (UP), India

#### ARTICLE INFO

##### Article History:

Received 18<sup>th</sup> November, 2016  
Received in revised form  
23<sup>rd</sup> December, 2016  
Accepted 15<sup>th</sup> January, 2017  
Published online 28<sup>th</sup> February, 2017

##### Key words:

Cooling chamber, Spray forming,  
Flight distance, Gas atomization plant.

#### ABSTRACT

The spray forming process is the process in which a molten metal is impinged by inert gases which results in atomized droplet. The resulting metal droplets while on flight and in residence time inside the spray cone cools rapidly before impinging onto the substrate. The temperature of smaller droplet drops down very fast and strikes the substrate in solid state while larger droplet strike the substrate in liquid form. Therefore to get the mushy droplet impinging on to the substrate knowledge of the thermal history of droplet is of utmost important. In order to determine the size of the cooling chamber of gas atomization plants, a fundamentally based mathematical model was described and the governing differential equation was solved using mathematical modelling method, and the relation between the flight distance and the time was established, then it is solved with the help of initial and boundary condition. With the help of the atomizer angle the size of the cooling chamber is calculated by the help of trigonometry.

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Citation: Pankaj Sharma, 2017. "Analysis and design of cooling chamber for gas atomization plant", *International Journal of Current Research*, 09, (02), 46614-46625.

## 1. INTRODUCTION

In recent years spray forming has been an emerging forming process for the production of near net shape products with the benefits that of rapid solidification, semi solid processing etc. This spray forming processes combine the advantage of metal casting and powder metallurgy. Spray forming has minimized the multiple steps of powder metallurgy which includes processes like powder production, sieving, de-gasing and consolidation into a single processing step and still micro-structural characteristics remains the same. Figure 1.1, illustrates the schematic view of spray forming. Professor Singer at the Swansea University first developed the idea of gas atomized spray forming in 1970s in which a high pressure gas jet impinges on a stable melt stream to cause atomization. The resulting droplets are then collected on a target, which can be manipulated within the sprays and used to form a near-dense billet of near-net shape. Spray forming, also known as spray casting, spray deposition is a method of casting near net shape metal components with homogeneous microstructures via the deposition of semi-solid sprayed droplets onto a shaped substrate. In spray forming an alloy is melted, normally in an induction furnace, then the molten metal is slowly poured into a conical tundish into a small-bore ceramic nozzle. The molten metal exits the furnace as a thin free-falling stream and is broken up into droplets by an annular array of gas jets, and these droplets then proceed

downwards, accelerated by the gas jets to impact onto a substrate. The process is arranged such that the droplets strike the substrate whilst in the semi-solid condition, this provides sufficient liquid fraction to 'stick' the solid fraction together. Deposition continues, gradually building up a spray formed billets of metal on the substrates. In the spray forming processes the metal is heated in the crucible until the superheat temperature is reached and the molten metal is poured in the tundish. The molten metal stream is poured into the atomization chamber using the gravity, where the molten metal stream gets disintegrated into spherical droplets due to jets of inert gases with very high kinetic energy. The spray thus formed gets accelerated towards the preformed substrate, cools down and solidifies partly as a result of high rate of heat transfer from the spray to the cold inert gas. The diameters of gas atomized droplets varies from 5 $\mu$ m to 500 $\mu$ m. Later on the droplets impacts on to the substrate, merges and forms the deposit. It was in 1960 in Swansea, Wales, by Singer and his colleagues when the first use of metal spray forming was used. In 1970s, spray forming was used as a substitute for conventional forming as production of perform was done directly from the melt. The spray forming process for money-making was first used by a number of singer's young researchers and as a result of which they founded the company Osprey Metals in Neath, Wales. Hence sometimes spray forming process is also called as the Osprey process. Since then, application potentials of the spray forming process has ignited several research and development works at universities and at various industries. In the late '80s Lavernia and Grant

\*Corresponding author: Pankaj Sharma,  
Department of Mechanical Engineering, Dr B R A College of Agril Engg & Tech, Etawah – 206001 (UP), India.

developed the liquid dynamic compaction (LDC) process which was similar to spray forming. LDC, Osprey process and spray formings are the generic names of similar or related processes. The purpose of the present work is to determine the sizes of the cooling chamber of gas atomized plants using tools and techniques of mathematical model and solving the mathematical equation by analytical method. The quality and the homogeneity of the microstructures of the deposit and its yield depend on the condition of droplets before impinging and hence it is of the utmost importance to determine and control the conditions and the temperature of the droplets.

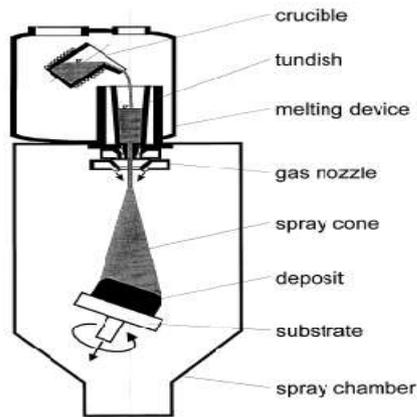


Fig 1.1. Schematic view of spray forming processes

## 2. Literature survey

The main intentions of this chapter is to deal with the previous works or researches carried out in the field of spray forming and gas atomized droplets. It has been found that the spray formed materials has many significance over conventional forming processes and the advantages are stated as under:-

- Absence of macro-segregation.
- Homogenous microstructures.
- Increased yield strengths.
- Decreased oxygen contaminants.
- Increased in workability and deformability.

### 2.1 Subdivision of spray forming processes

From the process technology view point spray forming is divided into many sub processes. The subdivision of the complete spray forming is shown in Fig 2.1.

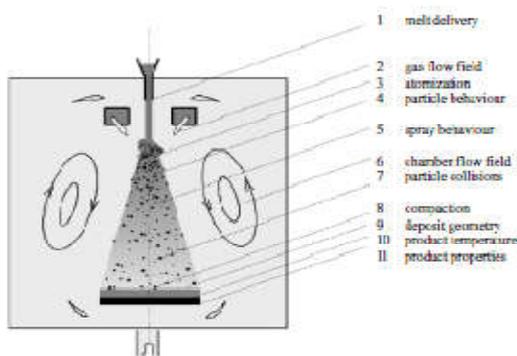


Fig 2.1. Sub process view of spray forming

The spray structure consists of in-flight accelerated, thus cooled and partially solidified, melt droplets as well as rapidly heated and decelerated gas flow. Analysis of individual droplet is done to know the behavior regarding movement and cooling of the droplet.

Ojha *et al.* 1992 described the reason of why analysis of droplet is done before they impinge onto the substrate and is shown as under:-

Table 2.1. Interaction of preform surface condition and spray condition in controlling the sticking efficiency

Condition of Deposition surface	Liquid	Melt Ejected due to low Viscosity	Melt Ejected due to low Viscosity	Melt Ejected due to low Viscosity
	Mushy	Partial bounce-off of the solid particles	Good Sticking	
	Solid	Total bounce-off of the solid particles	Partial sticking	Layered structure (e.g plasma spraying)
		Solid	Mushy	Liquid

spray condition

Interaction of preform surface condition and spray condition in controlling the sticking efficiency

Lawley *et al.* (1990) and Mathur *et al.* (1991) have inspected the spray forming process, and have discovered how fundamental knowledge of atomization and the compaction processes affect the system construction. In this way it was found that the appropriate control of processes parameters, such as substrate movements, sprays oscillation, deposit temperature and so on is must, as shown in Figure 2.2. This diagram consists of process that can be controlled by operator on the left side and processes that cannot be controlled by operator directly and the bottom consist of the spray conditions at impact and the surface conditions of the substrate/deposit.

The main purpose of Lawley *et al.*'s and Mathur *et al.*'s was to know parameter that can be controlled and they found that the significant parameters are

- Geometry and dimension of deposits
- The microstructures of the final product (porosity and grain size).

In his work, Ottosen (1993) identified the main function in analysis of an integral spray forming model was the modeling and simulation of complex heat transfer and momentum exchange processes.

Bauckhage and Uhlenwinkel (1996) laid emphasis on automated and optimized spray forming process, by dividing the spray formings in 3 parts which are melting and atomization, particles transport in spray and compaction.

The process parameters and product quality was linked by Payne *et al.* (1993) by the empirical spray forming process model. For suitable process control, Payne *et al.* has recognized:

- Process parameters controlled directly: e.g. spray time, melt temperature and GMR;
- Indirectly controllable process parameters: e.g. exhaust gas temperatures, deposit surfaces, roughness and porosity.

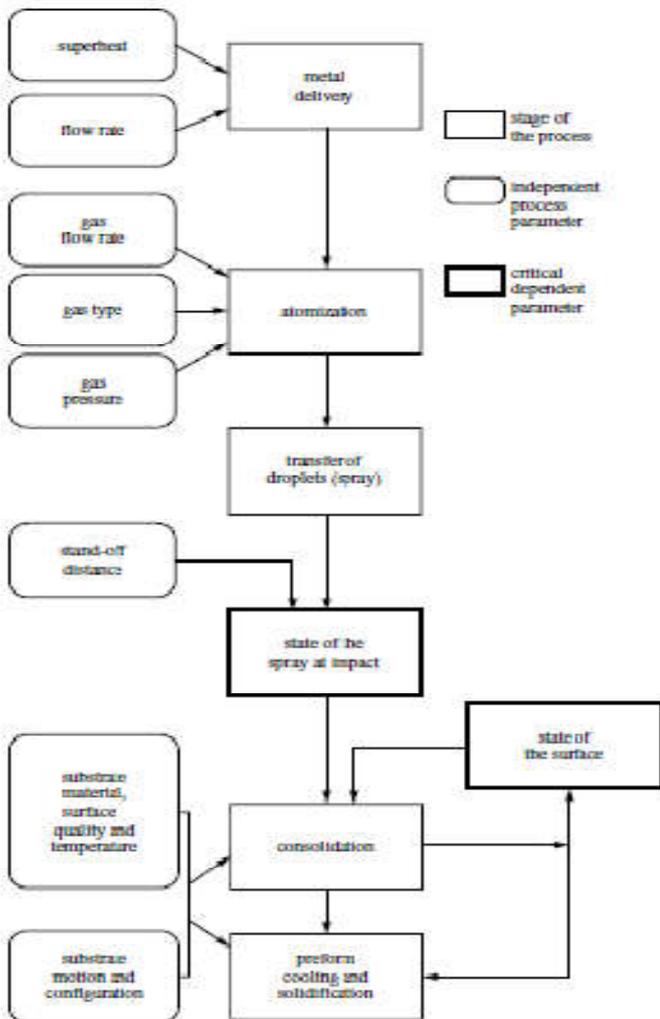


Fig. 2.2. Modelling of independent and dependent process parameters

The multi-coupled simulation of turbulent dispersed multiphase flow, containing gas as a continuous phase and droplets as a dispersed phase, is based on two modeling concepts:

- Eulerian/Lagrangians approach

This approach is related to direct intuitive approach which is applied in the analysis of the behavior of dispersed multiphase flow. In this individual particle is under the scanner of study and its interaction with local surroundings are analyzed on the scale of droplets size. Crowe *et al* (1977), Grant *et al* (1993), Bergmann *et al* (1995) were the researchers who published several models within spray formings application based on this approaches.

- Eulerian/Eulerians approach

In this the dispersed phase is considered to be as a quasi-second fluid with spatially averaged properties. Based on this approach derivation of the spray structure within the spray forming

process has been done by Liu (1990) and Fritsching *et al.* (1991).

2.2 Particle Movement

A fundamental description of the behavior of droplets in gas, the flow around gas atomized droplets and their analysis is given in Clift *et al* (1978), Crowe *et al* (1998), Sadhal *et al* (1997) and Sirignano (1999).

The various forces exerted on individual spherical particles are listed in Table 2.2.

Table 2.2. Various forces exerted on individual spherical particles

(1) inertial force	$F_i = -\rho_f V \frac{dv_f}{dt}$
(2) field force	$F_f = \rho_f V g$
• gravity force	
(3) pressure forces	$F_p = -\rho_f V g$
• buoyancy force	
• pressure gradients	$F_p = -\rho_f \frac{dv_k}{dt} V$
(4) fluid mechanics forces	
• resistance force	$F_w = -\frac{1}{2} \rho_f A_f c_w (Re)  v_f - v_k  (v_f - v_k)$
• added mass	$F_m = -\frac{1}{2} \rho_f V \left( \frac{dv_f}{dt} - \frac{dv_k}{dt} \right)$
(5) other forces	
• Basset history integral	$F_b = \frac{3}{2} d^2 \sqrt{\frac{\rho_f \mu}{\pi}} \left( \int_{-\infty}^t \frac{dv_f - dv_k}{(t-\tau)^{3/2}} d\tau + \frac{(v_f - v_k)}{\sqrt{t}} \right)$

The spherical droplet trajectory is derived from:  $\sum F=0$

The added-mass term describes the involvement of the surrounding gas, which gets accelerated together with the particle in the boundary layer of the particle. The last term of Basset history integral has been discovered by Reeks and McKee (1984) for the finite particle starting velocity.

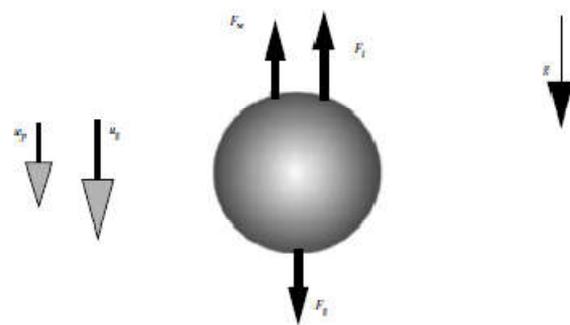


Fig. 2.3. Coordinate system for force-balancing for spherical droplet

In the analysis of gas atomized droplets, the density ratio of gas to the particles is negligible ( $\rho_g/\rho_p < 10^{-3}$ ). The particle trajectory equation can be simplified:

$$m_p \frac{du_p}{dt} = m_p g + \frac{1}{2} \rho_g |u_g - u_p| (u_g - u_p) c_d A_p \tag{2.1}$$

The force balance taken into account are force due to inertia, gravity and resistance. The resistance drag force coefficients  $c_d$  is described in the range of Reynolds number.

Re < 800

$$c_d = \frac{24}{Re} (1 + 0.15Re^{0.687}), \quad Re < 800. \tag{2.2}$$

Cliff *et al* (1978) found that in the area of Stokes flow  $Re < 1$ ,

$$c_d = \frac{24}{Re} \left( \frac{1 + \frac{2}{3}\overline{\mu}}{1 + \overline{\mu}} \right), \quad Re < 1. \tag{2.3}$$

Mahesh *et al* (2002) investigated the influences of dynamics of the droplets and temperature variations on the microstructures of final products. For this analytical models were constructed taking into consideration higher Reynolds number leading to supersonic flow of gases. The nozzles were designed so as to develop Mach no 3. Figure 2.4 describes the variation of velocity profile with flight distance. At first gas velocity gas substituted as 1000 m/s. and gas velocity had an inverse relation with the flight distance and was found to be 200 m/s at 0.7 m. We found the atomized gas velocity but taking a leap further. Instead of considering gas velocity as an average we opt to consider instantaneous velocity. Which would obviously be more appropriate?

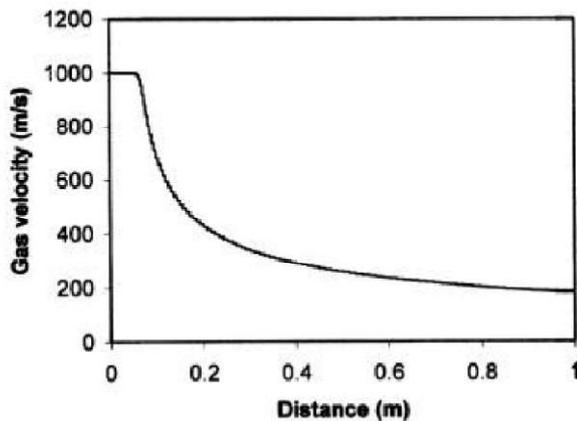


Fig. 2.4 Variation of gas velocity with respect to the flight distance

Fig. 2.4 shows the relation b/w flight distance, Reynolds's no., droplet size considering the velocity of the droplet relative to the gas. Reynolds's number is more predictable for the larger sized droplet.  $Re = \rho \cdot u \cdot d / \mu$

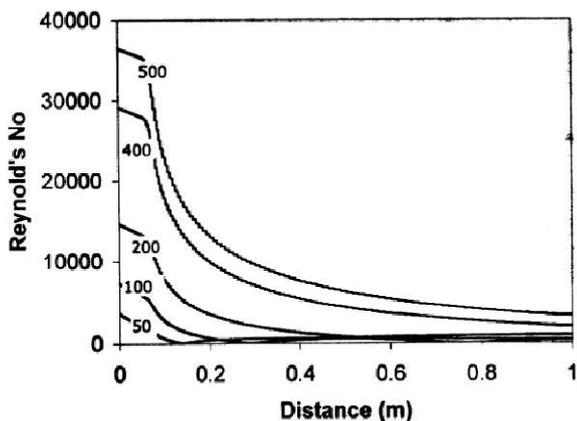


Fig 2.5. Variation of Reynolds number for different droplet size

Figure 2.6 shows the droplet's and gas velocity profile for varying sized droplets with respect to flight distance. It is obvious from the graph that at the exit of the nozzle the droplets have very less velocity (equal to acceleration due to gravity) and during the flight they gain velocity owing to the momentum transfer from the atomizing gases. The smallest droplet attains highest velocity during the flight and *vice versa*. The relative velocity of gas and droplet becomes zero as the flight distance increases and is equivalent to Reynold numbers variation.

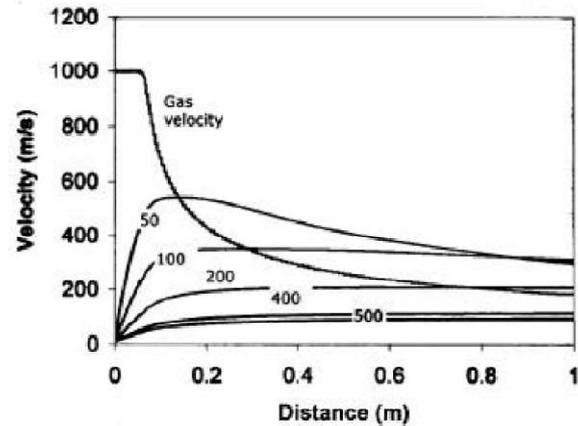


Fig. 2.6. Droplet's and Gas Velocity Profile for Varying Sized Droplets With respect to Flight Distances

The smaller droplets moves faster than atomizing gas and attain maximum velocity in minimum time. As the Reynolds number for smaller droplets are extremely small, counterintuitively large sized droplets had their drag coeff. Remaining more or less constant because of higher inertia where as for smaller droplets velocity varied considerably to their high inertia forces and thus no deceleration happens. Mostly droplets have a high velocity of the order of 100 m/s. the cooling mechanism of dendrites is explained on the basis of dendrite formation.

### 2.3 Heat transfer and Cooling of the droplet

The microstructure and the properties of sprayed alloy or of the final product can be known approximately by calculating the droplet thermal histories. The cooling due to convection is largely responsible for heat transfer in metal droplets because of a large temperature differences between molten metal droplets and cool atomizing gas. As a result for liquid metal droplets during atomization, convective cooling dominates over radiative cooling and hence radiation effect can be neglected (Lavernia *et al.*, 1988; Mathur *et al.*, 1989; Grant *et al.*, 1993; Eon-Sik Lee and Ahn, 1994). However, since the heat extractions from a droplets surface depends on the relative velocity between the cooling gas and the droplet itself, it is necessary to estimate droplet and gas velocities as discussed by Lavernia *et al.* (1988). Most of the researchers have adopted lumped parameter models (LPM) for calculation of heat transfer in gas atomized droplets (Lavernia *et al.*, 1988; Mathur *et al.*, 1988; Gutierrez *et al.*, 1989; Grant *et al.*, 1993; Eon-Sik Lee and Ahn 1994; Dimos and John, 1997). Moreover, Levi and Mehrabian (1982) shows that LPM give needed results when the temperature gradient inside the droplet is very large. The LPM is used for the simplicity of computation, since only first order ordinary differentials equations are to be solved. The small size of the droplets play a

significant role in neglecting the heat conduction within the droplets i.e. The droplet temperature is considered uniform (Lavernia *et al.*, 1988; Grant *et al.*, 1993; Eon/Sik Lee and Ahn, 1994). The process of conduction freezing using LPM as well as radially symmetric non-isothermal models have been analyzed by (Bayazitoglu and Cerny, 1993). In this not only the radial symmetry was imposed on the droplet but also the presence of recalescence resulting from severe undercoolings and phenomena of non-equilibrium was neglected. The LPM is accurate and also the assumption of uniform temperature inside the droplet is justified when the cooling rate was 104K/s. This follows the Newtonian cooling and gives rise to LPM.

**Equating the rate of change of sensible heat for the droplet to the rate of heat radiated from the surface. (Lavernia *et al.*, 1988),**

$$m_p c_p \frac{dT_p}{dt} = Nu \lambda_g \pi d_p (T_g - T_p) \tag{2.4}$$

Equation (2.4) is to be integrated. But heat transfer coefficients could not be considered to be a constant since the velocity of the gas decreases and that of the droplets increases. Neglecting radiation heat exchange and employing equation. (Ranz and Marshall, 1952)

$$Nu = 2 + 0.6 Re^{0.5} Pr^{0.33} \tag{2.5}$$

$$Pr = \frac{C_g \mu_g}{K_g} \tag{2.6}$$

Where,  $C_g$  is the gas specific heat,  $\mu_g$  the absolute viscosity of gas and  $K_g$  the gas thermal conductivity

Equation (2.5) has Nusslet no. that depends on the dia. Of the droplet and prandtl no. which is a function of the upstream flow conditions, reynold's no. here which is also a part of the eq. depends on the flow vel. Here (Lavernia *et al.*, 1988; Mathuret *et al.*, 1989; Grant *et al.*, 1993; Eon/Sik Lee and Ahn 1994) has limited validity (Dimos and John, 1997). As the Ranz and Marshall's correlation for 'h' is correct when the Reynolds number lies in the range of 0.1 to 4000. For supersonic gas atomization when Re. is more than 4000 Whitaker's (1972) correlation is used:

$$Nu_w = \frac{hD}{K} = 2 + (0.4 Re^{1/2} + 0.06 Re^{2/3}) Pr^{0.4} \left( \frac{\mu_\infty}{\mu_s} \right)^{1/4} \tag{2.7}$$

Heat transfer coefficient with the function of flight distance is to be known first before predicting the thermal states of droplets. For this Whitaker's correlation employed spherical surface considerations. From figure 7 it is evident that smaller droplets have higher coefficient of heat transfer due to higher surface area to volume ratio. The droplet history can be known Thereby

Figure 2.8 shows temperature variation for different droplet sizes as a function of flight distance. The solidus and liquidus temperature are illustrated to identify the physical state of the droplets based on their size when they reach the substrate. We know that mushy droplets provide best quality preform hence it

is important to know which droplet size strike the substrate in mushy zone and it was done by determining solid fraction of droplet.

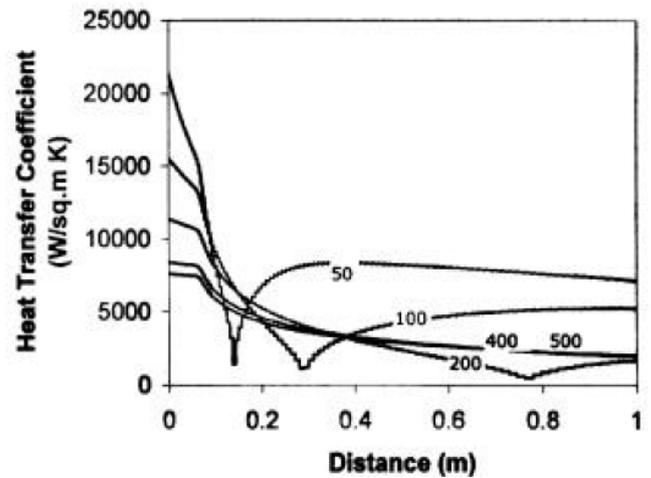


Fig. 2.7. The variations in heat transfer coefficients for different droplet size with the flight distance

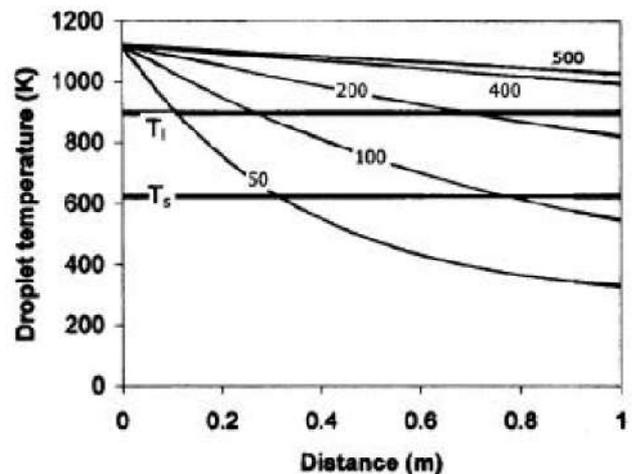


Fig. 2.8. Temperature variation for different droplet sizes as a function of flight distances

Figure 2.9 depicts percentage of solid in the droplets versus flight distance. It is seen from the figure that smaller droplets solidify completely at a distance from 0.2 to 0.6 m. The droplets of more than that of 100 μm sizes would be in semisolid/semi liquid state for a longer time. This analysis helps in optimizing the stand-off distances for given set of atomization parameters and for a particular metal and its alloys system. From this analysis it was found that the stand off distance of 0.6 to 0.7 m is suitable for obtaining preform in Al-Si-Mg alloy during the deposition trials. At this stand off distance it was anticipated that the droplets of size variety between 100 and 500 μm possess 90% to 15% solid fractions respectively. The parameters in spray casting be set up in such a way that the spray has more volume fraction of mushy droplets i.e. the droplets of size around 200–300 μm.

Figure 2.10 shows the rate of cooling in the droplets. From the figures it is evident that the cooling rate is very high for almost all the sizes of droplets. Cooling rate for smaller droplets were more as they lose heat faster.

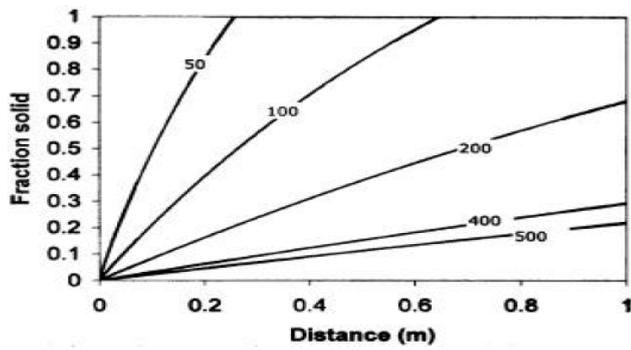


Fig. 2.9. Percentage of solid in the droplets versus flight distances

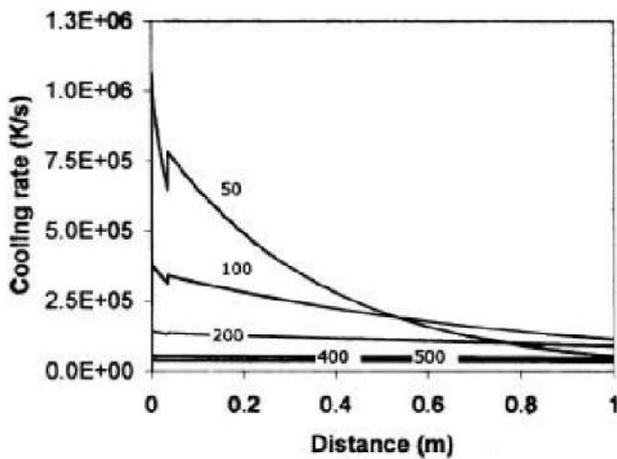


Fig 2.10. The rate of cooling in the droplets

2.4 Solidification of gas atomized droplets

Nucleation and growth of the crystal describes the solidification behavior of gas atomized droplets. Initially the solidification process is described by a homogeneous nucleation with slow cooling rates and thus without undercooling for pure metals. Here it is implicit that the superheated melt droplet while cooling to the phase change temperature releases the latent heat and transfers it across the surface. After solidification the particle mass cools down further. In the heterogeneous cooling model the foreign particles initiate the cooling process. In this model, upon reaching the solidification temperature, a balance exists between the released latent heat and the heat convectively transferred across the surface of the droplet and thus the temperature of the droplet remains constant during solidification. These solidification models have been used in spray forming, for example, by Zhang (1994) and Liu (1990). The solidification model described here (Bergmann, 2000) was developed for low carbon steel C30 (0.30 wt. % C), but may be easily adapted to other material compositions. The solidification model explained here is developed for low carbon steel C30 (0.30 weight% C) but is easily modified to other material compositions. Figure 2.10 shows part of the iron-carbon phase diagram, where the area for C30 is marked. For low cooling rates, temperature with respect to time curve can be drawn using this phase diagram. As in spray forming the cooling rate immediately after atomizations is very high hence there is a chance of undercooling even before the nucleation starts.

Starting with the superheated temperature  $T_m$ , the droplet cools down to liquidus temperature  $T_l$ . If the cooling rate is high

droplet may undercool until nucleation starts on reaching the nucleation temperature. As there is release in latent heat of fusion during recalescence, the droplet temperature increases until it reaches a local maximum in the cooling curve at  $T_r$ . Later on it follows segregated solidification, as the temperature keeps on decreasing. At temperature  $T_{per}$ , the peritectic transformation takes place at constant droplet temperature. As the peritectic transformation ends up, segregated solidification starts again until the droplet is completely solidified at  $T_s$ . After this in solidified state itself the droplet cooling continues.

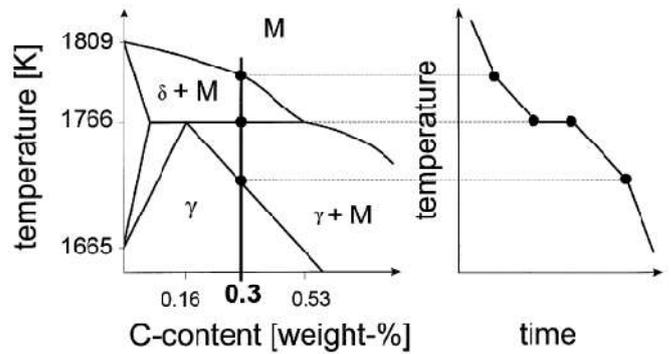


Fig 2.11. Phase Diagram of the Fe-C and the corresponding variation in phases with time

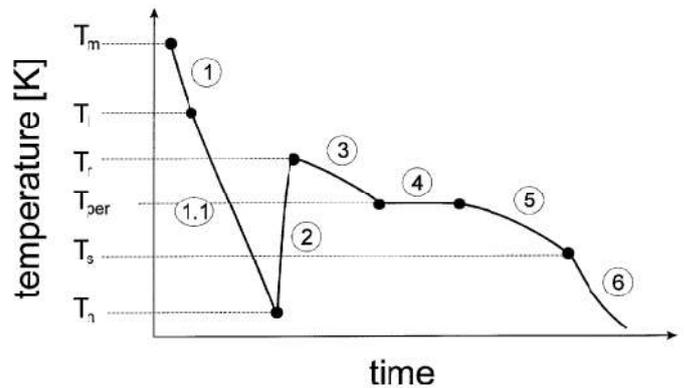


Fig 2.12. Temperature variation of gas atomized droplets with respect to time

Separate analysis of droplet cooling and solidification:

Cooling in the liquid state

For a spherical droplet, the change of internal heat content according to convection and radiation heat transfer can be expressed by:

$$c_{d,l} \frac{dT_d}{dt} = -\frac{6h}{\rho_d d_d} (T_d - T_g) - \frac{6\epsilon\sigma}{\rho_d d_d} (T_d^4 - T_w^4) \quad (2.8)$$

Where

$T_d$  = droplet temperature,  $T_g$  = gas temperature and  $T_w$  = temperature of the surrounding walls. The specific heat capacity of the liquid droplet material is  $c_{d,l}$ ;  $h$  is the heat transfer coefficient,  $\epsilon$  and  $\sigma$  are the emissivity and Stefan-Boltzmann constants,  $\rho_d$  and  $d_d$  are the droplet's density and diameter, respectively.

## Undercooling

The solidification process does not start immediately after liquidus temperature but the solidification depends on the cooling rate and on the size of the droplet. Nucleation temperature can be much lower than the liquidus temperature. The nucleation temperature for continuous cooling is defined as the temperature, where the number of nuclei  $N_n$  in the droplet volume  $V_d$  is identical to one:

$$N_n = V_d \int_{T_1}^{T_n} \frac{J(T)}{\dot{T}} dT = 1 \quad (2.9)$$

Heterogeneous nucleation minimizes the degree of undercooling. The maximum undercooling for iron based alloys is 295 Kelvin and a minimum undercooling of 3 Kelvin is assumed.

## Recalescence

As the solidification starts there is an increase in the temperature of droplet due to release of latent heat of fusion. The conservation equation for the droplet thermal energy is extended to:

$$c_d \frac{dT_d}{dt} = \Delta h_f \frac{df_s}{dt} - \frac{6h}{\rho_d d_d} (T_d - T_g) - \frac{6\varepsilon\sigma}{\rho_d d_d} (T_d^4 - T_w^4) \quad (2.10)$$

with  $f_s$  as fraction solid ( $f_s = 0$  droplet is completely liquid;  $f_s = 1$  droplet is completely solid) and the specific heat capacity of the droplets  $c_d$  as the average of the solid and liquid contents:

$$c_d = f_s c_{ds} + (1 - f_s) c_{dl} \quad (2.11)$$

The phase of recalescence ends, when the production rate of internal heat equals the heat transfer from the droplets surface. Here, the cooling curve of a droplets reaches a local maximum and the droplet temperature equals  $T_r$ :

$$\Delta h_f \frac{df_s}{dt} = \frac{6h}{\rho_d d_d} (T_r - T_g) + \frac{6\varepsilon\sigma}{\rho_d d_d} (T_r^4 - T_w^4) \quad (2.12)$$

## Segregated solidification 1

The heat conservation equation in this stage is described by

$$\begin{aligned} & \frac{dT_d}{dt} \left( c_d + \Delta h_f \frac{df_s}{dT_d} \right) \\ &= -\frac{6h}{\rho_d d_d} (T_d - T_g) - \frac{6\varepsilon\sigma}{\rho_d d_d} (T_d^4 - T_w^4) \end{aligned} \quad (2.13)$$

## Peritectic transformation

When the droplet temperature reaches the peritectic temperature, it remains at a constant value until this phase transformation got completed. The change in solid fraction during peritectic solidification is described by:

$$\Delta h_f \frac{df_s}{dt} = -\frac{6h}{\rho_d d_d} (T_d - T_g) - \frac{6\varepsilon\sigma}{\rho_d d_d} (T_d^4 - T_w^4) \quad (2.14)$$

Peritectic solidification gets completed, when the composition of the remaining liquid reaches the appropriate concentration.

## Segregated solidification 2

Segregated solidification again come into picture after peritectic transformation.

## Cooling in the solid state

Further cooling of droplet takes place after solidification. This process can be evaluated from the following equation:

$$c_{ds} \frac{dT_d}{dt} = -\frac{6h}{\rho_d d_d} (T_d - T_g) - \frac{6\varepsilon\sigma}{\rho_d d_d} (T_d^4 - T_w^4) \quad (2.15)$$

with  $c_{ds}$  as the specific heat capacity of the solid materials.

## Solidification behavior inside the melt particles

The temperature variation inside a single spherical droplet during solidification has been studied numerically by Kallien (1988) and Hartmann (1990). The simulation program was developed for solidification during metals casting. It includes undercooling thus calculates three-dimensional temperature variation in gas atomized droplet. The model is based on Fourier law for transient heat conduction in three-plane (Cartesian) coordinates as-

$$\rho c_p \frac{\partial T}{\partial t} = \frac{\partial}{\partial x} \left( \lambda \frac{\partial T}{\partial x} \right) + \frac{\partial}{\partial y} \left( \lambda \frac{\partial T}{\partial y} \right) + \frac{\partial}{\partial z} \left( \lambda \frac{\partial T}{\partial z} \right) \quad (2.16)$$

where conductivity  $\lambda$ , density  $\rho$  and heat capacity  $c_p$ , depend on location and temperature.

A modified temperature is introduced to achieve linear differential equation.

$$\Theta = \frac{1}{\lambda_1} \int_0^T \lambda dT \quad (2.17)$$

and thus linear differential equation was :

$$\frac{\rho c_p}{\lambda_0} \frac{\partial \Theta}{\partial t} = \frac{\partial^2 \Theta}{\partial x^2} + \frac{\partial^2 \Theta}{\partial y^2} + \frac{\partial^2 \Theta}{\partial z^2} \quad (2.18)$$

Finite difference method was used to solve this equation on an orthogonal-plane three dimensional grid.

The assumed boundary conditions are:

- The surrounding gas is assumed to be at constant temperature,
- Constant heat transfer coefficients across the whole surface of the droplet.
- At preselected nucleation temperature the solidification was initiated and nucleation was

considered to take place at either at single point or at number of grid cells.

A six stages approach for the particle solidification was assumed:

- (1) Cooling of the melts from superheating until the nucleation temperature is reached,
- (2) Attaining the highest undercoolings,
- (3) Solidification and recalescence,
- (4) Cooling and solidification in the melt temperature range between solidus and liquidus,
- (5) end of solidifications,
- (6) Cooling of the solidified particles.

For the recalescence phase, the releasing velocity of latent heat depend on undercooling  $\Delta T$ :

$$v = K\Delta T$$

As soon as the grid cells solidifies completely the adjacent cells begin to release the latent heat. Degree of undercooling controls the velocity of solidification. The heat transfer coefficient was taken as  $h = 20\,000 \text{ W/m}^2 \text{ K}$  and the undercooling prior to nucleations was considered to be 50 K. The solidification process initiates at a single point on the surface of the particle in a plane inside the particles. As there is release of latent heat, the interior of the particle gets heated up. For a 10% solidification rate, movement of the solidification front is visible, which raises the temperature of the surrounding grid cells close to the liquidus temperature. As Biot number is very less temperature variation inside the spherical droplets are neglected. Thus this behavior necessitates for refined modeling in order to obtain realistic spray forming modeling results.

### 3. Mathematical modelling

The modeling to determine the size of the gas atomized plant is done using fundamentally based mathematical model. Mathematical models are a set of trigonometry equations which represents a process or some aspect of it. Mathematical models have lots of merits over other modeling for e.g. no data is required, knowledge of mechanism of process is a must and there is no limit to these models as they can be applied anywhere.

#### 3.1 Development of fundamentally based mathematical model

- Preparation
- Mathematical formulation
  1. Assumptions
  2. The balance (force)
  3. Initial and boundary condition
- Solution of mathematical equations
  1. Analytical method
- Validation of model.

#### 3.2 Preparation

The parameter which is to be calculated is the size of the gas atomization plant which is a function radius  $r$ , and the atomizer angle. That means the complexity of the model is the two dimensional state. Here aluminium alloys are taken into consideration

**Table 3.1. Thermophysical properties of pure Aluminium**

Diameter of the droplets (dd)	20, 40, 60, 80, 100, 120, 140, 160 $\mu\text{m}$	Melting tem. ( $T_m$ )	933 K
Density ( $\rho_d$ )	2.707 $\text{g/cm}^3$	thermal diffusivity ( $\alpha$ )	$8.418 \times 10^{-5} \text{ m}^2/\text{s}$
Thermal conductivity ( $k_s$ )	204 $\text{W/m-K}$	Undercooling	50K

**Table 3.2. Properties of Nitrogen Gas**

Specific heat ( $C_{pg}$ )	1039 $\text{J/kg-K}$
Thermal conductivity ( $k_g$ )	$2.6 \times 10^{-2} \text{ W/m-K}$
Dynamic viscosity ( $\mu_g$ )	$1.78 \times 10^{-5} \text{ Ns/m}^2$
Density ( $\rho_g$ )	0.808 $\text{g/cm}^3$

### 3.3 Mathematical Formulation

#### 3.3.1 Assumption

- Property remains constant i.e., metal is homogenous and isotropic.
- No heat generation inside the spherical droplet.
- Surrounding gas atmosphere is assumed to be at constant temperature which is taken to be at 0 temperatures.
- The heat transfer coefficient is taken as constant across the whole surface of the particle.
- Homogeneous solidification occurs in the particle.
- There is no heat loss due to radiation.
- The Lumped Parameter Model used earlier by several researchers has not been taken into consideration.
- Droplets are considered to be perfectly spherical in shape with no deformation.
- There is no scattering or diffusion of the droplets.
- The added-mass term describes the involvement of the surrounding gas, which gets accelerated together with the particle in the boundary layer of the particle.

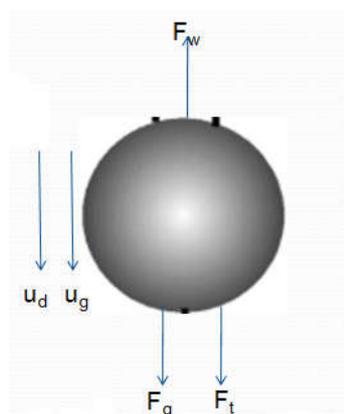
#### 3.3.2 The Balance

The spherical droplet trajectory is derived from:  $\Sigma F = ma$

$$m_p \frac{\partial u_p}{\partial t} = m_p g + \frac{1}{2} \rho_g I u_g - u_p I (u_g - u_p) C_d A_p$$

$$m_p \frac{\partial^2 h}{\partial t^2} = m_p g + \frac{1}{2} \rho_g I u_g - u_p I (u_g - u_p) C_d A_p$$

$$\text{where } A_p = \frac{\pi}{4} d^2$$



**Fig. 3.1. Balancing of forces on the droplet**

**3.4 Solution of mathematical equations**

$$C_d \text{ is given by } 0.28 + \frac{6\sqrt{Re} + 21}{Re}$$

Where Re is the reynold no. and is given by

$$Re = \frac{\rho g d u_g - u_p l}{\mu}$$

where  $\mu$  is the gas dynamic viscosity

**Table 3.3. For the calculation of the mass of the particles:**

Diameter of particle (d in $\mu\text{m}$ )	Density of particle ( $\rho_p$ in $\text{g/cm}^3$ )	Volume of particle ( $\text{cm}^3 * 10^{-9}$ ) $V_p = \frac{\pi}{6} d^3$	Mass of particle ( $m_p$ in $\text{gm} * 10^{-9}$ )
20	2.7	4.186667	11.2500
40	2.7	33.49333	90.4319
60	2.7	113.040	305.2080
80	2.7	267.9467	1412.9991
100	2.7	523.3333	1412.9991
120	2.7	904.320	2441.6640
140	2.7	1436.027	3877.2729

**Table 3.4**

**For evaluating reynold no.(Re)**

**Reynold number:** The Reynolds number is defined as the ratio of inertial forces to viscous forces and consequently quantifies the relative importance of these two types of forces for given flow conditions. It is given by:

$$Re = \frac{\rho g d u_g - u_p l}{\mu}$$

Density of particle ( $\rho_g$ in $\text{g/cm}^3$ )	Diameter of particle (d in $\mu\text{m}$ )	Velocity of gas ( $u_g$ in m/s)	Velocity of particle ( $u_p$ in m/s)	Gas dynamic viscosity ( $\mu$ in $\text{Pa-s} * 10^{-5}$ )	Reynold no. (Re $* 10^2$ )
0.808	20	980	228	3.78	3214.899
0.808	40	980	177	3.78	6865.862
0.808	60	980	150	3.78	10645.08
0.808	80	980	134	3.78	14467.05
0.808	100	980	122	3.78	18340.32
0.808	120	980	112	3.78	22264.89
0.808	140	980	105	3.78	3214.899

**Table 3.5**

**Calculation of the drag coefficients:**

**Drag coefficient:** In fluid dynamics, the drag coefficient (commonly denoted as  $c_d$ ) is a dimensionless quantity that is used to quantify the drag or resistance of an object in a fluid environment, such as atomizing gas or water.

$$C_d = 0.28 + \frac{6\sqrt{Re} + 21}{Re}$$

Reynold no.(Re)	Drag coefficients( $C_d$ )
321489.9	0.010648187
686586.2	0.00727208
1064508	0.005835353
1446705	0.00500311
1834032	0.004442053
2226489	0.004030622

**Table 3.6**

**Table for evaluating flight distance h:** The flight distance is calculated by the help of intial and boundry condition into the balance equation

$$m_p \frac{\partial U_p}{\partial t} = m_p g + \frac{1}{2} \rho_g l u_g - u_p l (u_g - u_p) C_d A_p$$

we know that  $\frac{\partial U_p}{\partial t} = \frac{\partial^2 h}{\partial t^2}$  so

$$m_p \frac{\partial^2 h}{\partial t^2} = m_p g + \frac{1}{2} \rho_g l u_g - u_p l (u_g - u_p) C_d A_p$$

by integrating and putting initial and boundry condition at  $t=0$ ,  $h=0$  and at  $t=0$ ,  $v = v_{max}$  that gives

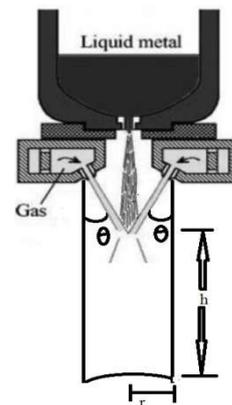
$$h = g t^2 + \left( \frac{1}{2} \rho_g (u_g - u_p)^2 C_d A_p t^2 \right) / m_p + v_{max} t$$

Mass of particle ( $m_p$ in gm)	Diameter of particle (d in $\mu\text{m}$ )	Cross sectional area ( $A_p$ in $\mu\text{m}^2$ )	Density of particle ( $\rho_g$ in $\text{g/cm}^3$ )	Velocity of gas ( $u_g$ in m/s)	Velocity of particle ( $u_p$ )	Flight distance (h in cm)
4186.667	20	314	0.808	980	228	15.33
33493.33	40	1256	0.808	980	177	28.8
113040	60	2826	0.808	980	150	35.26
267946.7	100	5024	0.808	980	134	40.94
523333.3	120	7850	0.808	980	122	44.1
904320	140	11304	0.808	980	112	46.28
1436027	160	15386	0.808	980	105	48.4

**3.4.5. Design of Cooling chamber for gas atomization plant:**

The radius of the cooling chamber can be determine by the help of Cooling chamber radius( $r$ ) =  $h * \tan \theta$

Where h is the flight distance and  $\theta$  is the atomizer angle



**Fig. 3.2. Design of cooling chamber for gas atomization plant**

**3.4.5 Calculation for the cooling chamber radius (r) with atomizer angle  $35^\circ$  ( $\theta=35^\circ$ )**

1. Radius ( $r_1$ ) =  $15.33 * \tan 35^\circ = 10.734$  cm
2. Radius ( $r_2$ ) =  $28.8 * \tan 35^\circ = 20.166$  cm
3. Radius ( $r_3$ ) =  $35.26 * \tan 35^\circ = 24.689$  cm
4. Radius ( $r_4$ ) =  $40.94 * \tan 35^\circ = 28.666$  cm
5. Radius ( $r_5$ ) =  $44.1 * \tan 35^\circ = 30.879$  cm
6. Radius ( $r_6$ ) =  $46.28 * \tan 35^\circ = 32.406$  cm
7. Radius ( $r_7$ ) =  $48.4 * \tan 35^\circ = 33.89$  cm

**3.4.6 Calculation for the cooling chamber radius (r) with atomizer angle  $37^\circ$  ( $\theta=37^\circ$ )**

1. Radius ( $r_1$ ) =  $15.33 * \tan 37^\circ = 11.552$  cm
2. Radius ( $r_2$ ) =  $28.8 * \tan 37^\circ = 21.702$  cm
3. Radius ( $r_3$ ) =  $35.26 * \tan 37^\circ = 26.57$  cm
4. Radius ( $r_4$ ) =  $40.94 * \tan 37^\circ = 30.85$  cm
5. Radius ( $r_5$ ) =  $44.1 * \tan 37^\circ = 33.23$  cm
6. Radius ( $r_6$ ) =  $46.28 * \tan 37^\circ = 34.874$  cm

7. Radius ( $r_7$ )=  $48.4 \cdot \tan 37^\circ = 36.47$  cm

### 3.4.6 Calculation for the cooling chamber radius (r) with atomizer angle $39^\circ$ ( $\theta=39^\circ$ )

1. Radius ( $r_1$ )=  $15.33 \cdot \tan 39^\circ = 12.414$  cm
2. Radius ( $r_2$ )=  $28.8 \cdot \tan 39^\circ = 23.321$  cm
3. Radius ( $r_3$ )=  $35.26 \cdot \tan 39^\circ = 28.552$  cm
4. Radius ( $r_4$ )=  $40.94 \cdot \tan 39^\circ = 33.153$  cm
5. Radius ( $r_5$ )=  $44.1 \cdot \tan 39^\circ = 35.711$  cm
6. Radius ( $r_6$ )=  $46.28 \cdot \tan 39^\circ = 37.477$  cm
7. Radius ( $r_7$ )=  $48.4 \cdot \tan 39^\circ = 39.194$  cm

### 3.4.7 Calculation for the cooling chamber radius (r) with atomizer angle $41^\circ$ ( $\theta=41^\circ$ )

1. Radius ( $r_1$ )=  $15.33 \cdot \tan 41^\circ = 13.326$  cm
2. Radius ( $r_2$ )=  $28.8 \cdot \tan 41^\circ = 25.035$  cm
3. Radius ( $r_3$ )=  $35.26 \cdot \tan 41^\circ = 30.651$  cm
4. Radius ( $r_4$ )=  $40.94 \cdot \tan 41^\circ = 35.589$  cm
5. Radius ( $r_5$ )=  $44.1 \cdot \tan 41^\circ = 38.336$  cm
6. Radius ( $r_6$ )=  $46.28 \cdot \tan 41^\circ = 40.231$  cm
7. Radius ( $r_7$ )=  $48.4 \cdot \tan 41^\circ = 42.073$  cm

### 3.4.8 Calculation for the cooling chamber radius (r) with atomizer angle $43^\circ$ ( $\theta=43^\circ$ )

1. Radius ( $r_1$ )=  $15.33 \cdot \tan 43^\circ = 14.295$  cm
2. Radius ( $r_2$ )=  $28.8 \cdot \tan 43^\circ = 26.856$  cm
3. Radius ( $r_3$ )=  $35.26 \cdot \tan 43^\circ = 32.880$  cm
4. Radius ( $r_4$ )=  $40.94 \cdot \tan 43^\circ = 38.177$  cm
5. Radius ( $r_5$ )=  $44.1 \cdot \tan 43^\circ = 41.124$  cm
6. Radius ( $r_6$ )=  $46.28 \cdot \tan 43^\circ = 43.156$  cm
7. Radius ( $r_7$ )=  $48.4 \cdot \tan 43^\circ = 45.134$  cm

### 3.4.9 Calculation for the cooling chamber radius (r) with atomizer angle $45^\circ$ ( $\theta=45^\circ$ )

1. Radius ( $r_1$ )=  $15.33 \cdot \tan 45^\circ = 15.330$  cm
2. Radius ( $r_2$ )=  $28.8 \cdot \tan 45^\circ = 28.800$  cm
3. Radius ( $r_3$ )=  $35.26 \cdot \tan 45^\circ = 35.260$  cm
4. Radius ( $r_4$ )=  $40.94 \cdot \tan 45^\circ = 40.940$  cm
5. Radius ( $r_5$ )=  $44.1 \cdot \tan 45^\circ = 44.100$  cm
6. Radius ( $r_6$ )=  $46.28 \cdot \tan 45^\circ = 46.280$  cm
7. Radius ( $r_7$ )=  $48.4 \cdot \tan 45^\circ = 48.400$  cm

## 3.5 Tables for cooling chamber sizes

### 3.5.1. For atomizer angle $35^\circ$ ( $\theta=35^\circ$ )

Flight distance(h in cm)	Cooling chamber size(r in cm)
15.33	10.734
28.8	20.166
35.26	24.689
40.94	28.666
44.1	30.879
46.28	32.406
48.4	33.89

### 3.5.2 For atomizer angle $37^\circ$ ( $\theta=37^\circ$ )

Flight distance (h in cm)	Cooling chamber size(r in cm)
15.33	11.552
28.8	21.702
35.26	26.57
40.94	30.85
44.1	33.23
46.28	34.874
48.4	36.47

### 3.5.3 For atomizer angle $39^\circ$ ( $\theta=39^\circ$ )

Flight distance(h in cm)	Cooling chamber size(r in cm)
15.33	12.414
28.8	23.321
35.26	28.552
40.94	33.153
44.1	35.711
46.28	37.477
48.4	39.194

### 3.5.4 For atomizer angle $41^\circ$ ( $\theta=41^\circ$ )

Flight distance(h in cm)	Cooling chamber size(r in cm)
15.33	13.326
28.8	25.035
35.26	30.651
40.94	35.589
44.1	38.336
46.28	40.231
48.4	42.073

### 3.5.5 For atomizer angle $43^\circ$ ( $\theta=43^\circ$ )

Flight distance(h in cm)	Cooling chamber size(r in cm)
15.33	14.295
28.8	26.856
35.26	32.880
40.94	38.177
44.1	41.124
46.28	43.156
48.4	45.134

### 3.5.6 For atomizer angle $45^\circ$ ( $\theta=45^\circ$ )

Flight distance(h in cm)	Cooling chamber size (r in cm)
15.33	15.330
28.8	28.800
35.26	35.260
40.94	40.940
44.1	44.100
46.28	46.280
48.4	48.400

## 4. Analysis and validation

It is obvious from the figure 4.1 that cooling chamber size varies with the respect of the atomizer angle and the flight distance. As Flight distance increases or atomizer angle increases, the size of the cooling chamber also increase and vice versa.

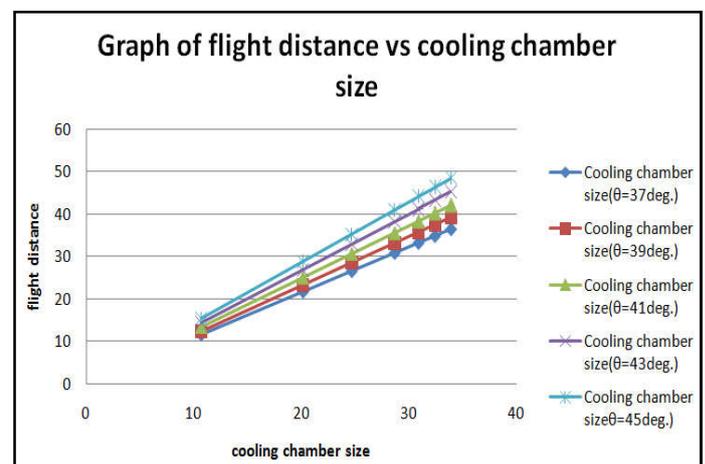


Figure 4.1

**Table 4.1**

The Table 4.1 shows the value of temperature for different droplet sizes as a function of time.

t (in sec)	Temp. in Kelvin forr=20 $\mu$ m	Temp. in Kelvin forr=40 $\mu$ m	Temp. in Kelvin forr=60 $\mu$ m	Temp. in Kelvin forr=80 $\mu$ m	Temp. in Kelvin forr=100 $\mu$ m
0	1147	1147	1147	1147	1147
0.004	660.4770328	744.6765096	826.2845	895.0959	913.1761
0.008	380.3225029	483.4726276	595.245	698.5149	727.0188
0.012	219.0011144	313.8890225	428.8071	545.1071	578.811
0.016	126.107416	203.7888245	308.9073	425.3907	460.8164
0.02	72.61643588	132.3075418	222.5329	331.9664	366.8758

Figure 4.2 shows the temperature variation in gas atomized droplets as a function of time.

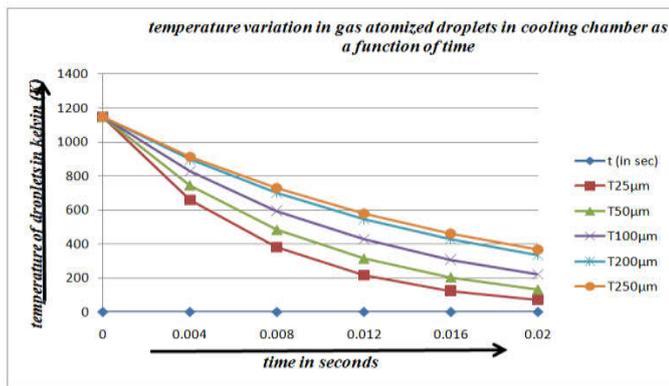
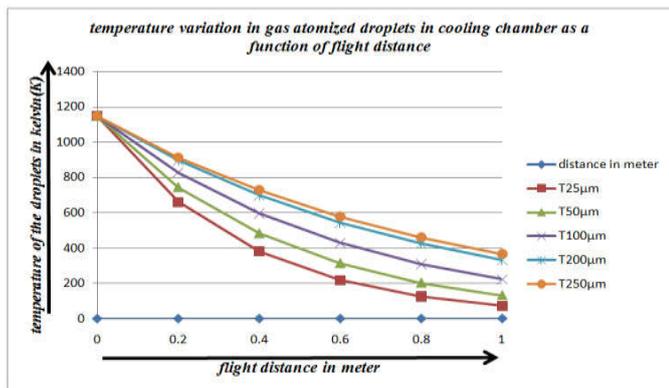
**Figure 4.2.**

Figure 4.3 shows the temperature variation in gas atomized droplets as a function of flight distance it is needed to calculate the cooling rate in K/s.

**Figure 4.3**

#### Validation of figure 4.2 and figure 4.3

Studying the figure 4.2 and from the past work done on the same topic we can say that the given graph can be validated. The temperature of the smaller droplets in cooling chamber decreases sharply as compare to bigger droplets because the area to volume ratio of smaller droplets is large and thus heat transfer rate is very fast in smaller droplets, where as larger droplets takes larger time to cool down.

#### Application of figure 4.2

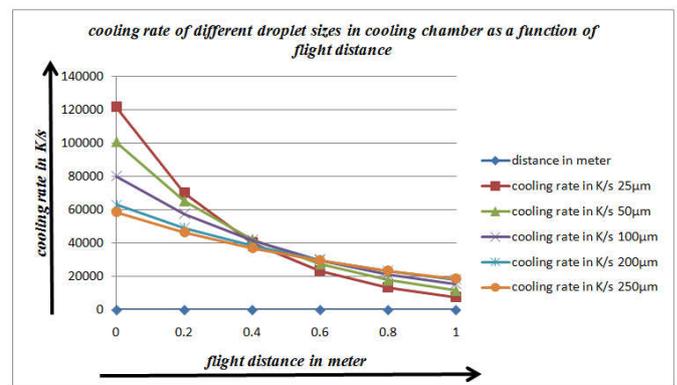
This graph can be used to determine that in cooling chamber what time and with what temperature droplets are striking the

substrate. This graph can also be used to set the initial velocity of atomization gas as it was found out from the literature survey it is the initial velocity of the atomization gas that determines the velocity of the droplets.

#### Application of figure 4.3

The graph has its own significance and application for examples using this graph it can be found out at how much distance what size of droplets remains in mushy stage as mushy stage of droplets is the first and the foremost requirements for best quality final product. This graph can also be used to determine the standoff distance, as it can be known using this graph the at what standoff distance how much percentage of droplets and in what range they remains in mushy stage or have solidified fraction between 10 to 90%.

Figure 4.4 shows the cooling rate of different droplet sized in terms of K/s as a function of flight distance.

**Figure 4.4**

#### Validation and application of Figure 4.4

It is obvious from the figure that cooling rate in cooling chamber for almost all droplets is of very high value. Although it can be noticed from the figure that initially the smaller droplets has the highest cooling rates in comparison to larger droplets but as the flight distance travelled by the droplets increases the cooling rate for smaller droplets decreases sharply and the cooling rates of larger droplets almost remains constant this is due to the fact that smaller droplets have already lost their heat during the initial stage of heat transfer. The application of cooling rate as a function of flight is to determine what should be the temperature of atomization gas in cooling chamber so that problem of rapid cooling or slow cooling can be solved to achieve the desired cooling rates and thus the desired state of droplets before impacting onto the substrate.

#### 5. Conclusion

- The size of the cooling chamber is directly proportional to the flight distance keeping atomizer angle constant.
- The size of the cooling chamber is directly proportional to the atomizer angle keeping keeping flight distance constant.
- The size of the cooling chamber is directly proportional to the size of the droplets.
- The size of the cooling chamber is highly influenced by the reynold number and drag coefficient.

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